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Luminescence of Eu^{3+} in $La_{1-x}Eu_xTa_7O_{19}$ (0 < x < 1) solid solution

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Abstract

The luminescence properties of Eu^{3+} in $La_{1-x}Eu_xTa_7O_{19}$ solid solution were examined systematically. In the structure the nearest neighbours of La^{3+} sites doped with Eu^{3+} ions are arranged two-dimensionally with proper separations of about 0.62 nm. An intense emission peak is observed at 610 nm together with supplementary weak peaks. All the peaks can be ascribed to ${}^5D_j \rightarrow {}^7F_{j'}$ (j=0,1; j'=0, 1, 2, 3, 4) transitions of Eu^{3+} . The emission of the ${}^5D_0 \rightarrow {}^7F_2$ transition becomes intense and saturates with the Eu^{3+} concentration. As a result, the profile of the concentration quenching implies that energy migration of excited Eu^{3+} hardly takes place in $La_{1-x}Eu_xTa_7O_{19}$ solid solution.

Keywords: Luminescence; Energy migration; Quenching

1. Introduction

Many investigations have been performed on the emission properties of concentrated Eu^{3+} -doped compounds, e.g. $EuMgB_5O_{10}$ [1], $Li_6Eu(BO_3)_3$ [2], EuP_3O_9 [3], NaEuTiO₄ [4], etc. These phosphors display a high critical concentration for quenching in comparison with conventional Eu^{3+} -doped compounds such as YAIO₃: Eu^{3+} [5]. This is due to the long separations and low dimensional arrangements of Eu^{3+} ions in La^{3+} sites [6,7].

Recently Gatehouse [8] demonstrated that $CeTa_7O_{19}$ consists of a layer of $(Ce^{3+}, Ta^{5+})-O^{2-}$ polyhedra interstratified in a double layer of $Ta^{5+}-O^{2-}$ polyhedra. Since $EuTa_7O_{19}$ is isomorphous with $CeTa_7O_{19}$, Eu^{3+} ions are separated by a shortest distance of about 0.62 nm and displaced two-dimensionally. In addition, each layer involves Eu^{3+} sites set about 1.0 nm apart.

The interesting luminescence properties of the rare earth polytantalates [9–12] have been examined extensively. Very recently, the present authors reported on the luminescence properties of $La_{1-x}Tm_xTa_7O_{19}$ [13] and $Y_{1-x}Tm_xTa_7O_{19}$ [14,15]. Most of the data indicated that the emission intensity changes peculiarly with the Tm^{3+} concentration. Ustimovich et al. [9] reported that an intense emission of Eu^{3+} is obtained at n=7 in the $(La_{1-x}Eu_x)_2O_3-nTa_2O_5$ system. However, the results were not sufficient to understand the luminescence properties of $La_{1-x}Eu_xTa_7O_{19}$ solid solution. As described above, the crystal structure involves the La^{3+} sites doped with Eu^{3+} . In addition, the separations between La^{3+} sites seem to be favourable to avoid the quenching phenomenon due to interaction between Eu^{3+} ions. In this paper the luminescence properties of $La_{1-x}Eu_xTa_7O_{19}$ are examined and discussed on the basis of the characteristics of activated Eu^{3+} sites.

2. Experimental details

Descriptions of the preparation of samples and the experimental set-up are given in Refs. [13,15].

3. Results and discussion

EuTa₇O₁₉ was prepared by the solid state reaction of Eu₂O₃ and Ta₂O₅ at 1200 °C in air. Fig. 1 shows a scanning electron microscope (SEM) photograph of the EuTa₇O₁₉ powder produced. The grains are 0.2–2 μ m in size and irregular in shape. This observation was independent of the Eu³⁺ concentration in La_{1-x}Eu_xTa₇O₁₉. From the powder X-ray diffraction patterns, all samples can be regarded as being single

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Fig. 1. SEM photograph of EuTa₇O₁₉.

phase. According to the data reported by Ustimovich et al. [9], $La_{1-x}Eu_xTa_7O_{19}$ is transformed from the LaTa₇O₁₉ structure type (orthorhombic) to the Eu-Ta₇O₁₉ structure type (tetragonal) at x = 0.05. However, we were able to index according to hexagonal symmetry for all Eu³⁺ concentrations, as reported in Refs. [16,17].

Fig. 2 shows the unit cell parameters *a* and *c* plotted against the *x* value of $La_{1-x}Eu_xTa_7O_{19}$. The unit cell parameters have a tendency to decrease linearly with increasing *x* because of the difference in ionic radius between La^{3+} (0.118 nm for eight coordination) and Eu^{3+} (0.107 nm) [18]. The result implies that Eu^{3+} ions are successively substituted for La^{3+} ions in the two-dimensional networks of $(La^{3+}, Ta^{5+})-O^{2-}$ polyhedra.

Fig. 3 shows the UV-visible diffuse reflection spectrum of EuTa₇O₁₉. It is seen that the absorbing edge is located around 270 nm (4.59 eV), which is similar to the value for LaTa₇O₁₉. The additional peaks can be ascribed to transitions from ${}^{7}F_{0,1}$ to each excited state of Eu³⁺ as indicated in the figure. There is a probability that a band due to charge transfer from O²⁻ to Eu³⁺ ions will be observed near 300 nm. However, no such band was observed in this spectrum.

Fig. 4 shows the emission spectra of La_{0.9}Eu_{0.1}Ta₇O₁₉ and EuTa₇O₁₉ under excitation at 396.3 nm measured at room temperature. The emission spectra consist of a main peak at 610 nm and several supplementary peaks. These peaks can be identified as ${}^{5}D_{j} \rightarrow {}^{7}F_{j'}$ (j = 0, 1; j' = 1, 2, 3, 4) transitions. Similar spectra were observed for all Eu³⁺ concentrations with some changes in intensity. In Fig. 4 the ${}^{5}D_{0} \rightarrow {}^{7}F_{2}$ (electric dipole transition)



Fig. 2. Concentration dependence of unit cell parameters of (a) a and (b) c of La_{1-x}Eu_xTa₇O₁₉.



Fig. 3. UV-visible diffuse reflection spectrum of EuTa₇O₁₉.



Fig. 4. Emission spectra of $La_{0.9}Eu_{0.1}Ta_7O_{19}$ and $EuTa_7O_{19}$ under excitation at 396.3 nm.

band is typically split into two or more bands. In comparison with the ${}^5D_0 \rightarrow {}^7F_1$ (magnetic dipole transition) band, the substitution sites for Eu³⁺ are considered to have no inversion symmetry. In addition, most powder samples were homogeneously coloured pale pink with increasing Eu³⁺ concentration. Consequently, the colouring must be caused by the Eu³⁺ emission itself.

Fig. 5 shows the emission spectra of $La_{0.9}Eu_{0.1}Ta_7O_{19}$ and $EuTa_7O_{19}$ upon cathode ray (CR) excitation under conditions of 15 kV and 0.5 μ A. These emission profiles are very similar to those of UV excitation. However, the intensity of the ⁵D₁ emission hardly decreases with increasing Eu³⁺ concentration. This implies that no cross-relaxation occurs between ⁵D₁ \rightarrow ⁵D₀ and ⁷F₀ \rightarrow ⁷F₃ transitions owing to the long separation of Eu³⁺ ions.

The Eu³⁺ concentration dependence of the emission intensity was examined using the ${}^{5}D_{0} \rightarrow {}^{7}F_{2}$ transition band at 610 nm under excitation at 396.3 nm in $La_{1-r}Eu_{r}Ta_{7}O_{19}$ as shown in Fig. 6. It can be seen that the observed plot deviates slightly from a proportional increase in emission intensity with Eu³⁺ concentration up to x = 1.0. Saturation of the emission intensity is frequently observed in phosphors with a high critical concentration. For instance, orthorhombic in $La_{1-r}Eu_rP_3O_9$ [3] the Eu^{3+} sublattice consists of zigzag chains with an intrachain separation between Eu³⁺ ions of 0.42 nm and an interchain separation of 0.73 nm. Therefore energy migration of excited Eu^{3+} is



Fig. 5. Emission spectra of $La_{0.9}Eu_{0.1}Ta_7O_{19}$ and $EuTa_7O_{19}$ upon CR excitation under conditions of 15 kV and 0.5 μ A.



Fig. 6. Concentration dependence of Eu^{3+} emission (${}^{5}D_{0} \rightarrow {}^{7}F_{2}$ transition) intensity under excitation at 396.3 nm in $La_{1-x}Eu_{x}Ta_{7}O_{19}$.

apparently predominantly to neighbouring sites displaced one-dimensionally. On the other hand, in $EuTa_7O_{19}$ the EuO_8^{13-} polyhedra are isolated from each other by TaO_7^{9-} and TaO_8^{11-} polyhedra [8]. Also, the shortest distance between Eu^{3+} ions is estimated to be about 0.62 nm. As a result, the separation of Eu^{3+} ions is too long for interaction of the exchange type to occur in $EuTa_7O_{19}$. Consequently, the interaction between Eu^{3+} ions is considered to be of the dipole-dipole type.

Buijs and Blasse [1,3] reported that the type and strength of Eu³⁺-Eu³⁺ interactions are fairly predictable from the arrangement of activated centres and the emission profile. The dipole-dipole interaction decreases abruptly with increasing separation of Eu³⁺ ions [6]. Energy migration to killer sites, resulting in concentration quenching, is restricted by the long separation and, as described in Ref. [7], the low dimensionality of the substituted Eu³⁺ sites. The dipole-dipole interaction of Eu³⁺ ions, where the strength depends on the oscillation strength of the ${}^{5}D_{0}-{}^{7}F_{0}$ transition, is closely related to the intensity of the ${}^{5}D_{0} \rightarrow {}^{7}F_{0}$ emission [3]. The ${}^{5}D_{0} \rightarrow {}^{7}F_{0}$ emission intensity is relatively weak in $La_{1-x}Eu_{x}Ta_{7}O_{19}$. It seems that this fact is another reason why the concentration quenching is very weak in $La_{1-x}Eu_{x}Ta_{7}O_{19}$.

4. Conclusions

The results of the Eu³⁺ concentration dependence of the emission intensity indicate that energy migration between Eu³⁺ ions hardly occurs in La_{1-x}Eu_xTa₇O₁₉. The long separation and low dimensionality of the La³⁺ sites doped with Eu³⁺ ions are eventually significant to avoid concentration quenching. Furthermore, the proper separation of Eu³⁺ results in no cross-relaxations. The emission spectrum of La_{1-x}Eu_xTa₇O₁₉ is independent of the concentration of Eu³⁺ in shape, even upon CR excitation. Therefore EuTa₇O₁₉ can be regarded as one of the concentrated Eu³⁺-doped compounds.

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